

Supplementary Information

Lithography-Free Patterning of SrTiO₃ -based Two-Dimensional Electron Gases using Direct Atomic Layer Processing

Anshu Gupta¹, Karolis Parfeniukas², Amit Chanda¹, Thor Hvid-Olsen¹, Mira Baraket²,

Maksym Plakhotnyuk², Kasper S. Pedersen³, Felix Trier^{1, *}

¹Department of Energy Conversion and Storage, Technical University of Denmark, Kgs. Lyngby, Denmark

²ATLANT 3D, Taastrup, Denmark

³Department of Chemistry, Technical University of Denmark, Kgs. Lyngby, Denmark

Email: fetri@dtu.dk

1. Extended structural information

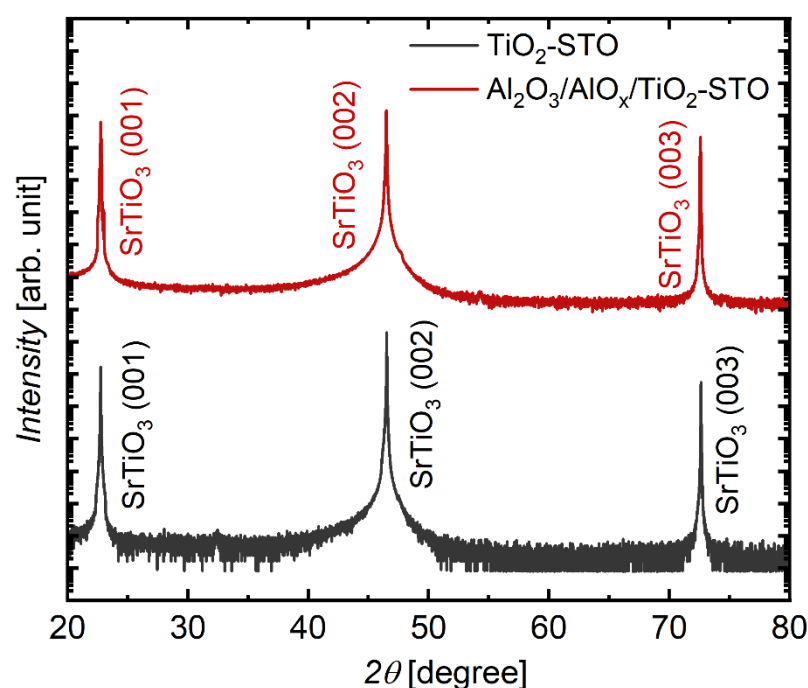


Figure S1. (a) X-ray diffraction (XRD) pattern of before deposition ($\text{TiO}_2\text{-STO}$) and after deposition ($\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-STO}$) at room temperature using a Rigaku SmartLab diffractometer

Alt text: XRD patterns comparing $\text{TiO}_2\text{-STO}$ substrate before deposition and $\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-STO}$ heterostructure after deposition, showing characteristic diffraction peaks and structural changes.

2. Extended morphological information

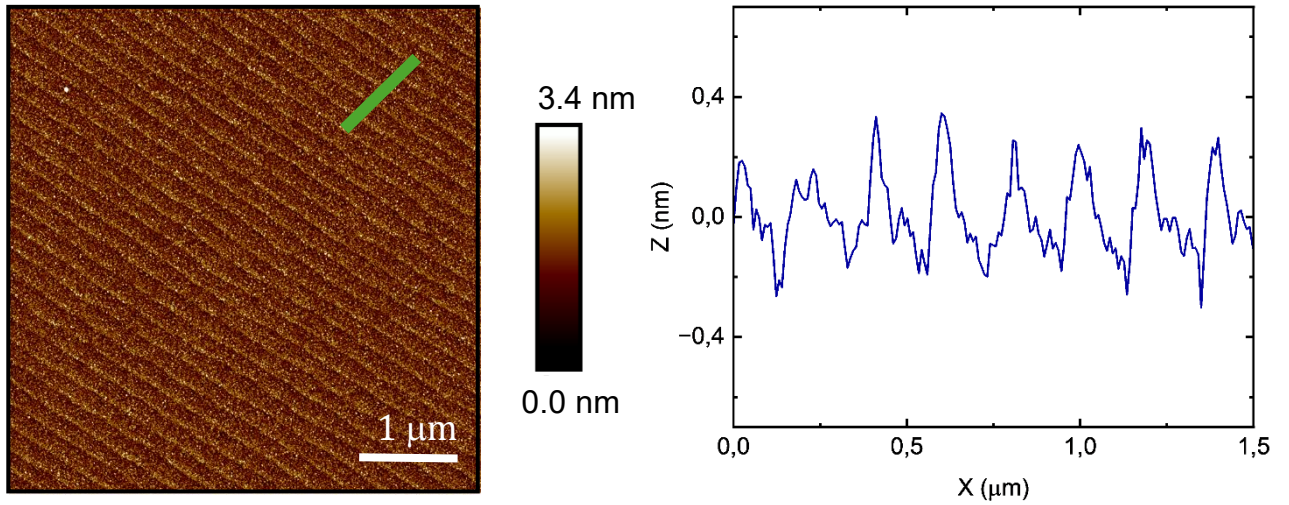


Figure S2. Atomic force microscopy image of the TiO₂-STO substrate showing root mean square (RMS) roughness approximately ≈ 0.3 nm. The image on the right highlights well-defined step-terrace features with a step height of about 0.4 nm.

Alt text: AFM images of TiO₂-STO substrate showing smooth surface morphology with low roughness and clear step-terrace structures.

3. Fitting results of the temperature dependent sheet resistance

	Parameters	Al ₂ O ₃ /AlO _x /TiO ₂ -STO
Low Temperature Fitting (2 K $\leq T \leq$ 25 K)	R_0 (Ω/sq)	617.78 ± 6.02
	R_{inel} ($\Omega \cdot \text{K}^{-n}/\text{sq}$)	0.123 ± 0.002
	n	2
	R_{Kondo} (Ω/sq)	49.13 ± 2.1
	T_K (K)	13.99 ± 1.5
	R ² value of the fit	0.9968

Table S1. List of parameters obtained by fitting the $R_s(T)$ data for Al₂O₃/AlO_x/TiO₂-STO heterostructure.

Alt text: Table listing fitting parameters extracted from temperature-dependent sheet resistance measurements of the Al₂O₃/AlO_x/TiO₂-STO heterostructure.

4. Optical microscopy images of $\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-SrTiO}_3$ patterned heterostructures and electrical measurement configuration.

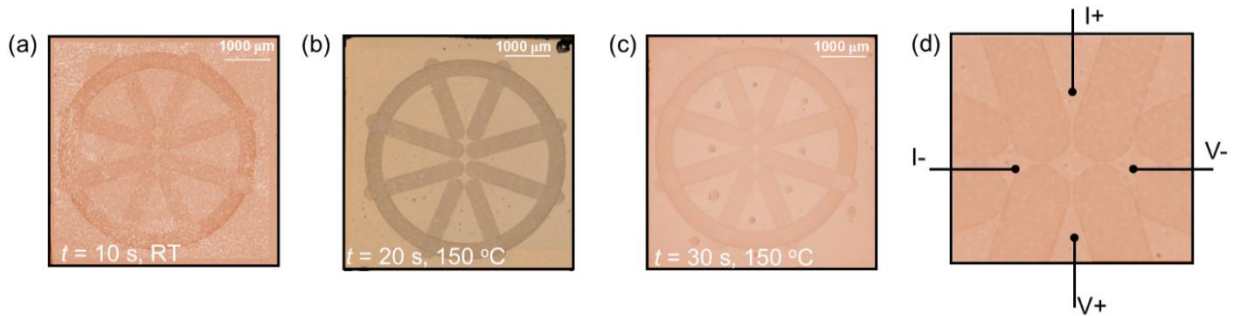


Figure S4: (a-c) Optical microscope images showing the $\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-SrTiO}_3$ heterostructure (STO) substrate after deposition. AlO_x deposition for (a) 10 s at room temperature 20°C , (b) 20 s at 150°C , and (c) 30 s at 150°C . The scale bar corresponds to $1000 \mu\text{m}$. (d) Schematic representation of the four-terminal electrical measurement configuration used for transport measurements, showing current ($I+/I-$) and voltage ($V+/V-$) contact arrangement.

Alt text: Optical microscope images of patterned $\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-SrTiO}_3$ heterostructures and transport measurement geometry. Panels (a-c) show circular TiO_2 patterned regions on STO substrates after AlO_x deposition under different conditions: (a) 10 s deposition at room temperature, (b) 20 s deposition at 150°C , and (c) 30 s deposition at 150°C . Each image includes a $1000 \mu\text{m}$ scale bar. Panel (d) illustrates the four-terminal measurement configuration, with current applied through the contacts ($I+$ and $I-$) and voltage measured across the contacts ($V+$ and $V-$).

5. Room-temperature (300 K) transport parameters of $\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-SrTiO}_3$

The table summarizes room-temperature (300 K) transport parameters of the $\text{Al}_2\text{O}_3/\text{AlO}_x/\text{TiO}_2\text{-SrTiO}_3$ heterostructures as a function of Al deposition time and growth temperature. The sheet resistance (R_s), carrier concentration (n_s), and Hall mobility (μ) show a strong dependence on deposition conditions, indicating tunable interfacial electronic reconstruction and carrier scattering with Al thickness and growth temperature.

t (s)	R_s (k Ω /sq.)	n_s (cm ⁻²)	μ (cm ² /Vs)
10 s, $T_G = 150$ °C	12.0	7.7×10^{13}	6.7
20 s, $T_G = 150$ °C	27.8	5.7×10^{14}	0.4
30 s, $T_G = 150$ °C	28.6	2.0×10^{14}	1.0
10 s, $T_G = 20$ °C	50.4	1.3×10^{14}	0.9

Table S2: Room-temperature (300 K) transport parameters of Al₂O₃/AlO_x/TiO₂-SrTiO₃ heterostructures for different Al deposition times (t) and growth temperatures (T_G). The sheet resistance (R_s), sheet carrier density (n_s), and Hall mobility (μ) are extracted from Hall measurements, showing systematic variation with deposition conditions.

Alt text: Room-temperature Hall transport parameters of Al₂O₃/AlO_x/TiO₂-SrTiO₃ heterostructures, including sheet resistance, carrier density, and mobility as a function of Al deposition time and growth temperature.

6. Sheet resistance R_s dependence on the Al deposition time

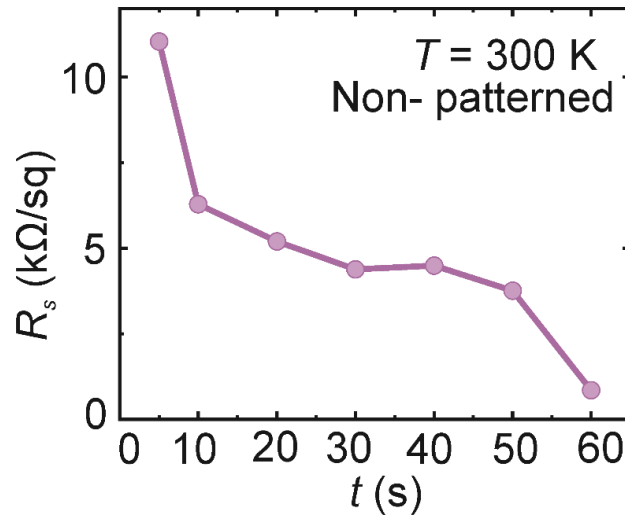


Figure S5: Sheet resistance R_s dependence on the Al deposition time at $T = 300$ K. The data were acquired for samples without TiO₂ patterning.

Alt text: Figure presents the dependence of R_s measured at 300 K as a function of Al deposition time during sputtering of samples prepared without TiO₂ mask. R_s decreases systematically with increasing deposition time, corresponding to a greater Al thickness. Specifically, R_s decreases from approximately ≈ 20.27 k Ω /sq. for a 5 s deposition to about ≈ 1.55 k Ω /sq. for a 60 s deposition.

7. Symmetrization and antisymmetrization of magnetotransport data

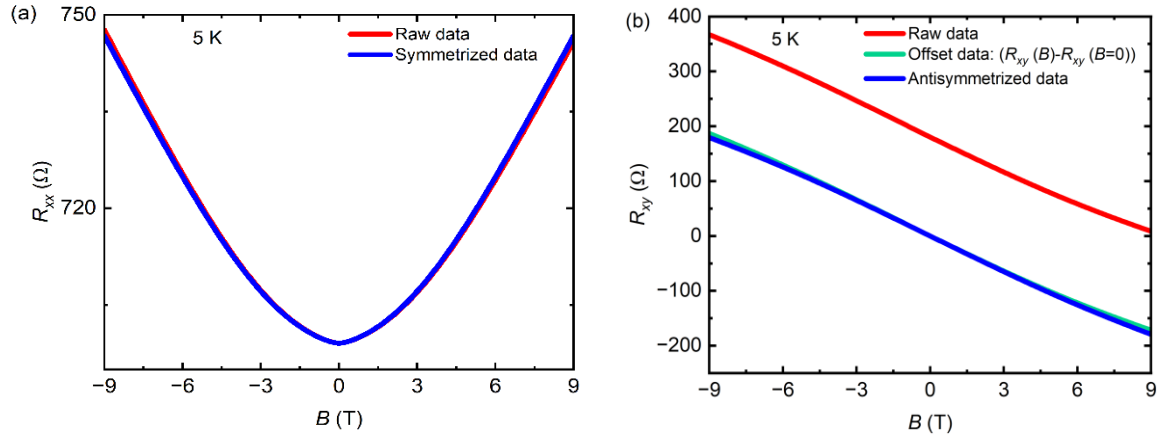


Figure S6: (a) Raw longitudinal resistance $R_{xx}(B)$ and symmetrized magnetoresistance obtained (b) Raw Hall resistance $R_{xy}(B)$, offset Hall resistance and antisymmetrized Hall resistance extracted. The procedure removes the contribution arising from contact misalignment and isolates the longitudinal and transverse transport components.

Alt text: The left panel presents symmetrized magnetoresistance as a function of magnetic field at low temperature, showing a symmetric field dependence. The right panel shows antisymmetrized Hall resistance versus magnetic field, after removing symmetric contributions, and the offset Hall resistance obtained by subtracting the raw Hall data $R_{xy}(B)$ from Hall resistance at zero magnetic field ($R_{xy}(B = 0)$).

8. Time-Dependent Evolution of Electrical Transport Properties

We have compared the transport properties measured shortly after growth with those obtained now after storage time in a 1 bar N_2 environment for over 250 days. We did not observe any significant long-term degradation or systematic evaluation of the transport properties in our sample. As shown here, the sheet resistance at 5 K changes from $\approx 699 \Omega/\text{sq.}$ (time = 0 days) to $667 \Omega/\text{sq.}$ (time = 250 days), while at 300 K it changes from approximately $11.5 \text{ k}\Omega/\text{sq.}$ to $12.0 \text{ k}\Omega/\text{sq.}$ The corresponding Hall analysis shows only minor variations in carrier density and mobility. At 5 K, the mobility changes from $\approx 734.6 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ (time = 0 days) to $895.4 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ (time = 250 days), while the total carrier density n_s changes from $\approx 5.5 \times 10^{13} \text{ cm}^{-2}$ (time =

0 days) to $6.2 \times 10^{13} \text{ cm}^{-2}$ (time = 250 days). In our case, the conducting interface is stabilized by the 33 nm thick Al_2O_3 capping layer, resulting in stable transport properties over time.

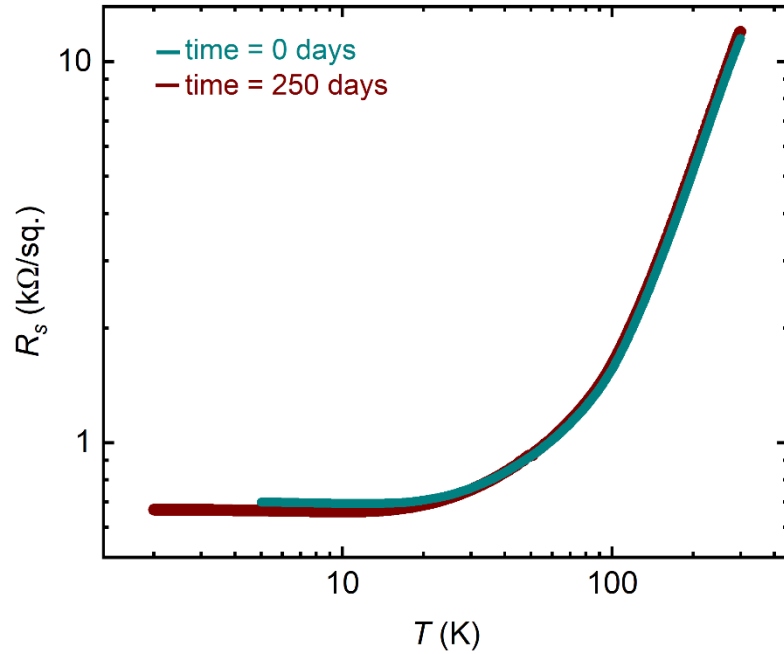


Figure S7: Temperature dependence of sheet resistance, $R_s(T)$, measured between 5 K and 300 K immediately after growth ($t = 0$ days) and after 250 days of aging, showing the evolution of transport behavior over time.

Alt text: Temperature-dependent sheet resistance of the sample measured from 5 K to 300 K immediately after growth and after 250 days of storage, showing similar transport behavior over time.